

## Synthesis and Structure of a New High-Pressure Form of $\text{Rh}_2\text{O}_3$ \*

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A new high-pressure form of  $\text{Rh}_2\text{O}_3$  has been prepared at 65 kbar and 1200°C. Four-probe resistivity measurements made on a single crystal show  $\text{Rh}_2\text{O}_3$  II to be a semiconductor with a room-temperature resistivity of 130  $\Omega$  cm and an activation energy of 0.16 eV.

$\text{Rh}_2\text{O}_3$  II is orthorhombic with  $a = 5.1686 \pm 3$ ,  $b = 5.3814 \pm 4$ ,  $c = 7.2426 \pm 4$  Å,  $Z = 4$ ,  $D_{\text{obsd}} = 8.28$  gm/cc,  $D_c = 8.36$  gm/cc, and space group,  $Pbna$ . Although the unit cell is similar to that of an orthorhombic perovskite, the structure, refined by least squares, is similar to that of corundum in that it contains pairs of  $\text{RhO}_6$  octahedra which share faces. However, only two edges of each octahedra are shared with other octahedra. The structure may also be described as containing layers of the corundum structure cut parallel to (10 $\bar{1}$ 1), and stacked together to form the  $\text{Rh}_2\text{O}_3$  II structure.

In the study of the crystal structures of  $\text{ABO}_3$  phases, much attention has been given to criteria for the formation of the corundum and perovskite structures. Recently, Megaw (1) discussed the similarities between the lithium niobate structure, which is directly related to that of corundum, and that of perovskite. In this note, we report a new high-pressure phase of rhodium sesquioxide,  $\text{Rh}_2\text{O}_3$  II, which has a unit cell similar to that of orthorhombic perovskite, but a structure related to that of corundum.

In these experiments, finely divided  $\text{Rh}_2\text{O}_3$  produced by heating  $\text{RhCl}_3$  in air at 800°C was placed in a Pt capsule and heated at 1200–1500°C for 1 hr at 65 kbar and quenched rapidly. Oxygen analysis of the resultant metallic-gray crystals of  $\text{Rh}_2\text{O}_3$  II by a catalytic fusion method indicated 19% O. The calculated oxygen content of  $\text{Rh}_2\text{O}_3$  is 18.9%. Resistivity measurements using a four-probe technique (2) showed a room-temperature resistivity of 130  $\Omega$  cm and an activation energy of 0.16 eV.

An X-ray diffraction pattern taken using a Hägg-Guinier camera with a KCl internal standard is

given in Table I. The unit cell dimensions, refined by a least-squares technique, are  $a = 5.1686 \pm 3$ ;  $b = 5.3814 \pm 4$ ; and  $c = 7.2426 \pm 4$  Å. The calculated density, 8.36 gm/cc, compares reasonably well to the measured density, 8.28 gm/cc. Comparison of the unit cell volume of 201.45 Å<sup>3</sup> for 4-formula units to 208.03 Å<sup>3</sup> for ambient-pressure, low-temperature rhombohedral  $\text{Rh}_2\text{O}_3$  (3) and 205.56 Å<sup>3</sup> for the ambient-pressure, high-temperature, orthorhombic  $\text{Rh}_2\text{O}_3$  reported by Wold et al. (3), indicates percentage volume changes  $\Delta V/V_1$  of 3.2% and 2.0%, respectively. The space group determined by precession photographs of a single crystal was found to be  $Pbna-D_{2h}^{14}$ , in contrast to the space group  $Pbnm-D_{2h}^{16}$  found for the orthorhombic perovskite  $\text{GdFeO}_3$  (4, 5).

Reid and Ringwood (6) recently predicted the existence of a perovskite-type  $\text{Fe}_2\text{O}_3$  at pressures of 600–1200 kbar in shock wave experiments by McQueen and Marsh (7). Reid and Ringwood based their conclusion on an extrapolated zero-pressure density of 5.96 gm/cm<sup>3</sup>, which is 14% higher than that of  $\alpha$ - $\text{Fe}_2\text{O}_3$  (hematite). We originally thought that the  $\text{Rh}_2\text{O}_3$  system might be analogous to the  $\text{Fe}_2\text{O}_3$  system since both  $\text{Rh}_2\text{O}_3$  I and hematite have the corundum structure and the high-pressure phases appeared to have the orthorhombic perovskite structure. However, we were suspicious of the large difference between the density changes in the

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TABLE I  
POWDER DIFFRACTION PATTERN OF Rh<sub>2</sub>O<sub>3</sub> II

Index	$d_{\text{obsd}}$	$d_{\text{calcd}}$	$I_{\text{obsd}}$
0 0 2	3.621	3.621	20
1 1 1	3.314	3.314	30
0 2 0	2.691	2.691	70
1 1 2	2.598	2.597	100
2 0 0	2.584	2.584	70
0 2 1	2.521	2.522	5
2 1 0	2.330	2.330	5
1 2 1	2.266	2.267	5
1 0 3	2.187	2.187	5
0 2 2	2.161	2.160	2
2 0 2	2.103	2.104	5
1 1 3	2.027	2.027	5
1 2 2	1.990	1.993	2
2 2 0	1.864	1.864	80
0 0 4	1.811	1.811	30
2 2 1	1.805	1.805	10
0 2 3	1.797	1.797	5
2 2 2	1.657	1.657	10
1 3 1	1.650	1.650	40
1 1 4	1.629	1.629	30
3 1 1	1.600	1.600	5
1 3 2	1.535	1.534	60
0 2 4	1.502	1.502	20
3 1 2	1.494	1.495	80
2 0 4	1.483	1.483	30
2 2 3	1.475	1.475	10
1 3 3	1.387	1.387	5
1 1 5	1.350	1.350	2
0 4 0	1.345	1.345	5
0 4 1	1.322	1.323	5
2 2 4	1.299	1.299	25
4 0 0	1.292	1.292	10

two systems. Using criteria established by Shannon and Prewitt (8) relating density changes with changes in coordination, it appeared that Rh<sub>2</sub>O<sub>3</sub> II should contain Rh in six-coordination, whereas the high-pressure Fe<sub>2</sub>O<sub>3</sub> probably contains Fe in greater than six-coordination.

To resolve this problem, the crystal structure of Rh<sub>2</sub>O<sub>3</sub> II was solved using observed structure factors obtained with a Picker single-crystal diffractometer, MoK $\alpha$  X-rays, and a spherically ground Rh<sub>2</sub>O<sub>3</sub> II crystal of 0.008-cm radius. Full-matrix least-squares refinement resulted in a discrepancy index  $R(R = \sum |F_0| - |F_c| / \sum |F_0|)$  of 0.042. The final atom parameters and interatomic distances are given in Table II.

The Rh<sub>2</sub>O<sub>3</sub> II structure is related to that of corundum in that it contains pairs of RhO<sub>6</sub> octahedra which share faces. However, each octahedron

TABLE II  
FINAL ATOM PARAMETERS AND INTERATOMIC DISTANCES FOR Rh<sub>2</sub>O<sub>3</sub> II<sup>a</sup>

	$x$	$y$	$z$	$B$
Rh	0.7498(1)	0.0312(2)	0.1058(1)	0.15(4)
O(1)	0.6037(11)	0.1161(10)	0.8494(8)	0.08(8)
O(2)	0.0505(16)	$\frac{1}{2}$	0	0.14(12)
		Rh-O(1,1) 00 $\bar{1}$	2.055 Å	
		-O(1,3) 0 $\bar{1}$ 0	2.076	
		-O(1,4) 000	2.068	
		-O(1,5) 0 $\bar{1}$ 0	2.017	
		-O(2,1) 100	2.094	
		-O(2,5) 0 $\bar{1}$ 0	1.985	
		Rh-O(av.)	2.050	
		-O(1,5) 1 $\bar{1}$ 0	3.449	
		-O(1,6) 100	3.171	
		-O(1,8) 0 $\bar{1}$ 1	3.431	
		-O(2,5) 0 $\bar{1}$ 0	3.255	

<sup>a</sup> The second number within the parentheses for each oxygen atom indicates the symmetry transformation applied to the coordinates of this atom. The numbers,  $pqr$ , just to the right of the parentheses are the translations applied to the transformed coordinates. For example, 0(2,5) 0 $\bar{1}$ 0 means that symmetry transformation 5 was applied to 0(2) and then -1 was subtracted from the transformed  $y$  coordinate. The symmetry transformations are  $xyz$ ;  $\frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2} - z$ ;  $\frac{1}{2} - x, \bar{y}, \frac{1}{2} - z$ ;  $x, \frac{1}{2} - y, \bar{z}$ ;  $\bar{x}, \bar{y}, \bar{z}$ ;  $-\frac{1}{2} + x, -\frac{1}{2} - y, -\frac{1}{2} + z$ ;  $-\frac{1}{2} + x, y, -\frac{1}{2} - z$ ;  $-\frac{1}{2} + y, z$ .

shares only two edges with other octahedra rather than three as in corundum. In fact, the structure may be thought of as containing slabs of the corundum structure cut parallel to (10 $\bar{1}$ 1) which, incidentally, is the most common twinning plane of corundum crystals. These slabs stacked together form the Rh<sub>2</sub>O<sub>3</sub> II structure. Figure 1a and b are schematic representations showing the relation between the two structures. Figure 1a is the Rh<sub>2</sub>O<sub>3</sub> II structure projected down the  $a$  axis. Two unit cells are shown here with only the bonds from the rhodium atoms (small circles) to the oxygen atoms in the shared octahedral face drawn in. The dashed lines from the rhodium atom in the left center of the figure indicate the other three oxygen atoms required to complete the octahedron. The dark motifs represent Rh<sub>2</sub>O<sub>3</sub> units centered around  $x = \frac{1}{2}$ , and the light ones  $x = 0$ . Figure 1b shows Rh<sub>2</sub>O<sub>3</sub> with the corundum structure drawn to emphasize the similar aspects of the two structures. The drawing is projected down the hexagonal  $a$  and the direction of the hexagonal  $c$  is indicated by the Rh-Rh vector

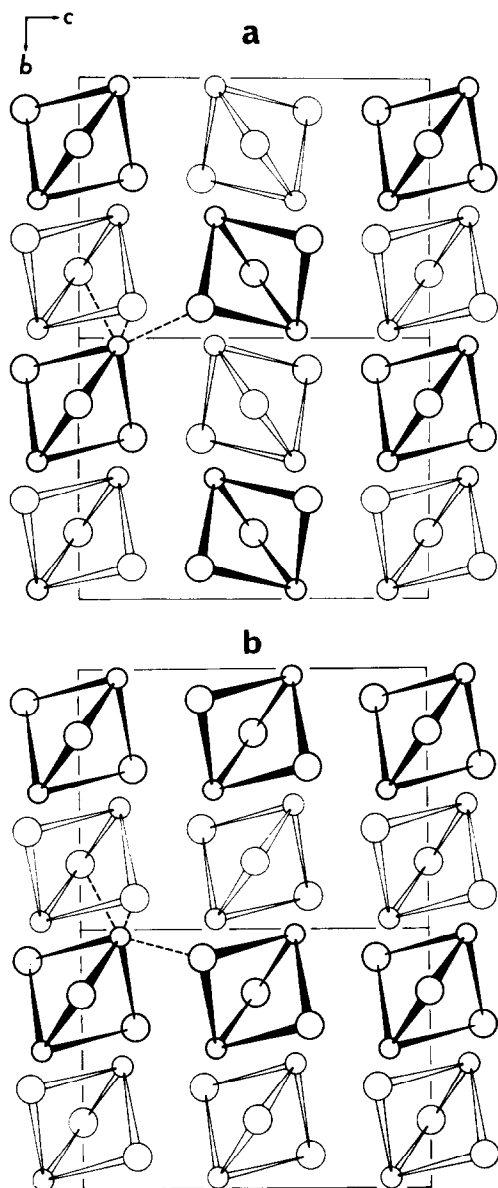


FIG. 1. (a) The  $\text{Rh}_2\text{O}_3$  II structure projected along the  $a$  axis. The small circles are rhodium atoms and the dark motifs are centered around  $x = \frac{1}{2}$  and the light ones around  $x = 0$ .

(b) The  $\text{Rh}_2\text{O}_3$  I structure (corundum) projected along the hexagonal  $a$ . The hexagonal  $c$  is defined by the vector between rhodium pairs sharing a common octahedral face.

across the shared octahedral faces. In both figures the columns on the left and right of each drawing are essentially identical. The center columns are related by a twofold axis at  $\frac{1}{2}, y, \frac{1}{2}$  in Fig. 1a. That is, if this operation is applied to the center column of the structure in Fig. 1a, the structure in Fig. 1b will result. Note that five of the six oxygen

atoms coordinating Rh remain the same, but the sixth is different. This causes the greater distortion of the octahedra in the corundum-type structure.

The average Rh–O distance predicted for six-coordinated  $\text{Rh}^{3+}$  using effective ionic radii (9) is 2.045 Å. From Table II the observed average is seen to be 2.050 Å. The Rh–Rh distance is 2.80 Å, somewhat shorter than the 2.90 Å predicted for  $\text{Rh}_2\text{O}_3$  I by Prewitt, Shannon, Rogers, and Sleight (10).

### Summary and Conclusions

A new form of  $\text{Rh}_2\text{O}_3$  has been synthesized which has a structure similar to that of corundum but a diffraction pattern much like that of orthorhombic perovskite. Although the new structure is denser than the  $\text{Rh}_2\text{O}_3$  corundum structure, the oxygen atoms are not obviously more perfectly close packed than in the corundum structure although the  $\text{RhO}_6$  octahedra are less distorted.

It is possible that ilmenite and lithium niobate equivalents of the  $\text{Rh}_2\text{O}_3$  II structure will be found and that possibly some other phases previously thought to be orthorhombic perovskite actually have this structure.

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### References

1. H. D. MEGAW, *Acta Cryst.* **A24**, 583 (1968).
2. T. A. BITHER, J. L. GILLSON, AND H. S. YOUNG, *Inorg. Chem.* **5**, 1559 (1966).
3. A. WOLD, R. J. ARNOTT, AND W. J. CROFT, *Inorg. Chem.* **2**, 972 (1963).
4. S. GELLER, *J. Chem. Phys.* **24**, 1236 (1956).
5. P. COPPENS AND M. EIBSCHÜTZ, *Acta Cryst.* **19**, 524 (1965).
6. A. F. REID AND A. E. RINGWOOD, *J. Geophys. Res.* **74**, 3238 (1969).
7. R. G. MCQUEEN AND S. P. MARSH, in "Handbook of Physical Constants," pp. 153–159 (S. P. Clark, Ed.), Memoir 97 of the Geological Society of America, Inc. Revised Edition, 1966.
8. R. D. SHANNON AND C. T. PREWITT, *Mater. Res. Bull.* **4**, 57 (1969).
9. R. D. SHANNON AND C. T. PREWITT, *Acta Cryst.* **B25**, 925 (1969).
10. C. T. PREWITT, R. D. SHANNON, D. B. ROGERS, AND A. W. SLEIGHT, *Inorg. Chem.* **8**, 1985 (1969).